

# Basella alba rubra spinach pigment-sensitized TiO<sub>2</sub> thin film-based solar cells

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**Abstract** Nanocrystalline TiO<sub>2</sub> thin films have been prepared by sol–gel dip coating method. The X-ray diffraction results showed that TiO<sub>2</sub> thin films annealed at 400, 450 and 500 °C are of anatase phase and the peak corresponding to the (101) plane is present in all the samples. The grain size of TiO<sub>2</sub> thin films was found to increase with increasing annealing temperature. The grain size is found to be 20, 25 and 33 nm for the films annealed at 400, 450 and 500 °C. The structure of the TiO<sub>2</sub> nanocrystalline thin films have been examined by high-resolution transmission electron microscope, Raman spectroscopy and FTIR spectroscopy. TiO<sub>2</sub> thin films were sensitized by natural dyes extracted from basella alba rubra spinach. It was found that the absorption peak of basella alba rubra extract is at about 665 nm. The dye-sensitized TiO<sub>2</sub>-based solar cell sensitized using basella alba rubra exhibited a  $J_{sc}$  of 4.35 mA cm<sup>-2</sup>,  $V_{oc}$  of 0.48 V, FF of 0.35 and efficiency of 0.70 %. Natural dyes as sensitizers for dye-sensitized solar cells are promising because of their environmental friendliness, low-cost production and fully biodegradable.

**Keywords** Sol–gel · TiO<sub>2</sub> nanocrystalline thin film · Dye-sensitized solar cell

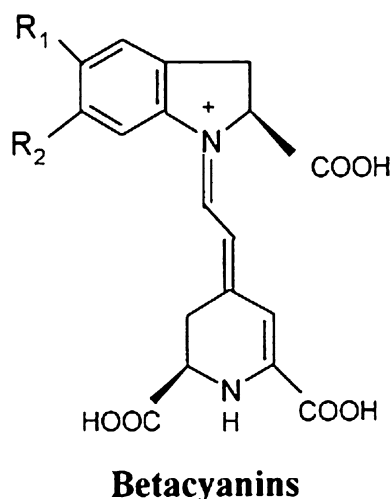
## Introduction

Solar energy conversion based on dye sensitization of wide band gap nanocrystalline semiconductor films is an area of intense investigation (O'Regan and Gratzel 1991; Gratzel 2001; Hagfeldt et al. 1994). The most efficient dye-sensitized solar cells are based on ruthenium-centered polypyridyl metal organic complexes because of their strong absorption of visible light, favorable spatial separation of HOMO and LUMO states and because they can be repetitively oxidized and reduced without degradation (Nazeeruddin et al. 1993). Much of the research on natural sensitizers has focused on either chlorophyll or anthocyanin pigments. These have typically resulted in energy conversion efficiencies of <1 % (Cherepy et al. 1997; Dai and Rabani 2002; Kumara et al. 2006; Wongcharee et al. 2007; Polo and Murakami Iha 2006). The highest conversion efficiency for a dye-sensitized solar cell sensitized with a natural dye was 1.49 % obtained using chlorophyll-containing extracts from *Rhoeo spathacea* Stearn (Lai et al. 2008). Recently, 5.4 % conversion efficiency has been obtained by co-adsorption of two synthetic dyes obtained from chlorophyll precursors (Wang et al. 2010) which were extracted from plants. Another class of plant pigments with great potential for solar energy conversion is the betalains, consisting of the red betacyanins and yellow betaxanthins. Betalains are thought to serve the same functions in plants as anthocyanins, acting as “sunscreens” and antioxidants (Stintzing and Carle 2004; Heuer et al. 1994; Castellar et al. 2008). Recently, betacyanin class of plant pigments has been used in a dye-sensitized solar cell in which red

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**Fig. 1** The chemical structure of betacyanins molecules

beet root pigments were employed and energy conversion efficiency of 0.67 % has been reported (Zhang et al. 2008). The reddish-purple pigment betacyanin (Fig. 1) is commonly found in beets, bougainvillea flowers, prickly pear, etc., in most cases coexisting with another yellow or orange betaxanthin dye (Heuer et al. 1994; Castellar et al. 2008). *Basella alba* (family Basellaceae) known as Malabar spinach or cyclone spinach is a fast-growing vegetable, native of tropical Asia, probably originating from India or Indonesia. *Basella alba rubra* contains maximum betacyanin flavonoid. In this paper, betacyanin pigment extracted from *basella alba rubra* leaves are used to sensitize TiO<sub>2</sub> nanocrystalline thin films for dye-sensitized solar cells. A recent calculation shows that the visible transition of betacyanin is well-described as a HOMO → LUMO excitation which moves electron density from the aromatic ring to the dihydropyridyl moiety (Qin and Clark 2007).

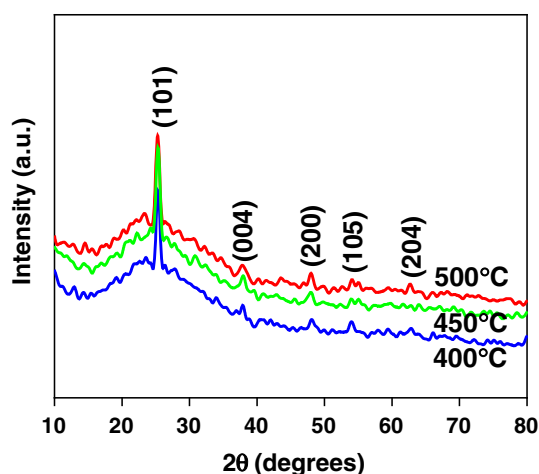
## Experimental

Fresh *basella alba rubra* spinach was cut into very small pieces and then soaked in 100 ml of ethanol at room temperature for 24 h and the solid residues were filtered out. The ethanol fraction was separated and few drops of concentrated HCl were added so that the solution became deep red in color (pH < 1). Then, the dye solution was stored at 4 °C before use. To prepare the photo-anode of dye-sensitized solar cells, the ITO conducting glass sheet (Asahi Glass; Indium-doped SnO<sub>2</sub>, sheet resistance: 15 U/square) was first cleaned in a detergent solution using an ultrasonic bath for 15 min, rinsed with double-distilled water and then dried. The matrix sol was prepared by mixing 1.1 ml of titanium isopropoxide with 15 ml of isopropanol at room temperature and stirred for half an

hour. Then 0.22 ml of glacial acetic acid is added drop wise and stirred vigorously for 2 h to obtain a homogeneous mixture of TiO<sub>2</sub> sol. The prepared sol was deposited on the ITO glass by sol–gel dip coating method. The film was dried at 80 °C for 30 min in air and then annealed at 400, 450 and 500 °C in a muffle furnace. The *basella alba rubra* spinach dye solution was used for sensitizing TiO<sub>2</sub> electrodes. The structural properties of the films have been studied using X-ray diffraction method (Rigaku Rint 2000 series). High-resolution transmission electron microscope (HRTEM) images of the prepared TiO<sub>2</sub> have been recorded using a JEOL JEM 2100 microscope. The absorption spectra of the films have been recorded using spectrophotometer (Jasco V-570). Lithium iodide, iodine and acetonitrile purchased from Sigma Aldrich have been used as received for the preparation of electrolyte. The redox electrolyte with [I<sub>3</sub><sup>-</sup>]/[I<sup>-</sup>] 1:9 was prepared by dissolving 0.5 M LiI and 0.05 M I<sub>2</sub> in acetonitrile solvent. Since LiI is extremely hygroscopic, electrolytes were prepared in a dry room maintained at dew point of 60 °C. The counter electrode was prepared using platinum chloride as follows: the H<sub>2</sub>PtCl<sub>6</sub> solution in isopropanol (2 mg ml<sup>-1</sup>) was deposited onto the ITO glass by spin coating method. TiO<sub>2</sub> film annealed at 400, 450 and 500 °C has been used for the fabrication of solar cell. TiO<sub>2</sub> electrode was immersed in the extracted dye solution at room temperature for 24 h in the dark. The electrode was then rinsed with ethanol to remove the excess dye present in the electrode and then the electrode was dried. The counter electrode was placed on the top of the TiO<sub>2</sub> electrode, such that the conductive side of the counter electrode faces the TiO<sub>2</sub> film with a spacer separating the two electrodes. The two electrodes were clamped firmly together using a binder clip. Now, the prepared liquid electrolyte solution was injected into the space between the clamped electrodes. The electrolyte enters into the cell by capillary action. This resulted in the formation of sandwich-type cell. Natural dye-sensitized TiO<sub>2</sub>-based solar cells have been fabricated with area of nearly 0.25 cm<sup>2</sup>, and it was found that the cell efficiency was independent of cell area in this range as reported by Gokilamani et al. (2013). The J–V characteristic of the cell was recorded using a Keithley 4200-SCS meter. A xenon lamp source (Oriel, USA) with an irradiance of 100 mW cm<sup>-2</sup> was used to illuminate the solar cell (equivalent to AM1.5 irradiation).

## Results and discussion

TiO<sub>2</sub> nanocrystalline films have three well-known phases namely: anatase, rutile, and brookite. Rutile and anatase are tetragonal where as brookite is orthorhombic. Rutile is the only stable phase. Anatase and brookite are metastable at



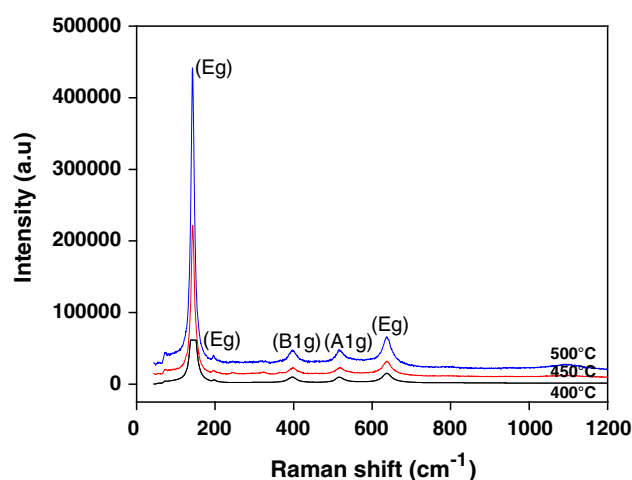
**Fig. 2** X-ray diffraction pattern of TiO<sub>2</sub> thin films for different temperatures

all temperatures and can be converted to rutile after heat treatment at high temperatures (Lee et al. 2007). Figure 2 shows the diffraction pattern of the sol-gel prepared TiO<sub>2</sub> films, annealed at 400, 450 and 500 °C. A narrow peak at 25.35° corresponding to (101) reflection of the anatase phase of TiO<sub>2</sub> has been observed in the diffraction pattern (JCPDS No. 21-1272). The grain size has been calculated using Scherrer's formula (Culty 1978)

$$D = \frac{k\lambda}{\beta \cos \theta}$$

where  $D$  is the grain size,  $k$  is a constant taken to be 0.94,  $\lambda$  is the wavelength of the X-ray radiation,  $\beta$  is the full width at half maximum and  $\theta$  is the angle of diffraction. The average grain size was found to be 20, 25 and 33 nm for the films annealed at 400, 450 and 500 °C, respectively. Grain size is found to increase with increase in annealing temperature. The annealing temperature facilitates the subsequent crystal growth process, accompanied by the diffusion of titania species forming big-sized anatase crystals and causing the merge of some adjacent mesopores. At the same time, the spatial confinement by mesopore arrays controls the formation and growth of anatase phase, leading to a more or less uniform distribution of titania nanocrystals (Que et al. 2006).

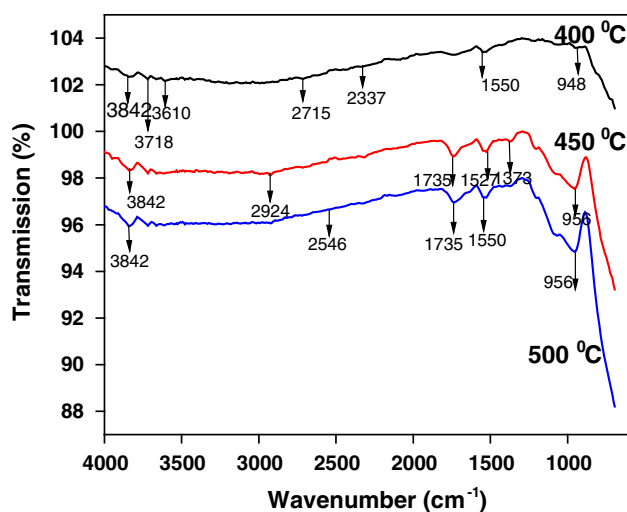
To discriminate the local order characteristics of the TiO<sub>2</sub> films, we carried out non-resonant Raman spectroscopy studies. The technique is non-destructive, capable to elucidate the titania structural complexity as peaks from each crystalline phase are clearly separated in frequency, and therefore the anatase and rutile phases are easily distinguishable. Figure 3 shows the Raman spectra of the nanocrystalline TiO<sub>2</sub> sample annealed at 400, 450 and 500 °C. Raman peaks originate from the vibration of molecular bonds, that is vibrational mode Eg and A1g peaks, which are



**Fig. 3** The Raman spectra of TiO<sub>2</sub> thin film for different temperatures

related to different crystal planes. According to factor group analysis, anatase has six Raman active modes (A1g + 2B1g + 3Eg). Ohsaka (1980) have reported the Raman spectra of an anatase TiO<sub>2</sub> and have stated that six allowed modes appear at 144 cm<sup>-1</sup> (Eg), 197 cm<sup>-1</sup> (Eg), 399 cm<sup>-1</sup> (B1g), 513 cm<sup>-1</sup> (A1g), 519 cm<sup>-1</sup> (B1g), and 639 cm<sup>-1</sup> (Eg). The vibrational peaks of the nanocrystalline TiO<sub>2</sub> samples at 143 cm<sup>-1</sup>, 197 cm<sup>-1</sup>, 396 cm<sup>-1</sup>, 519 cm<sup>-1</sup>, 638 cm<sup>-1</sup> and the absence of overlapped broad peaks show that the material is well crystallized, with low number of imperfect sites. These peaks are unambiguously attributed to the anatase modification. A special attention must be given to the Raman peaks observed at 143 cm<sup>-1</sup>, 197 cm<sup>-1</sup>, 396 cm<sup>-1</sup>, 519 cm<sup>-1</sup>, 638 cm<sup>-1</sup> which are all slightly shifted, probably due to a smaller size of TiO<sub>2</sub> nanocrystalline particles (Arabatzis et al. 2002). There was no much shift in the peaks due to increase in temperature.

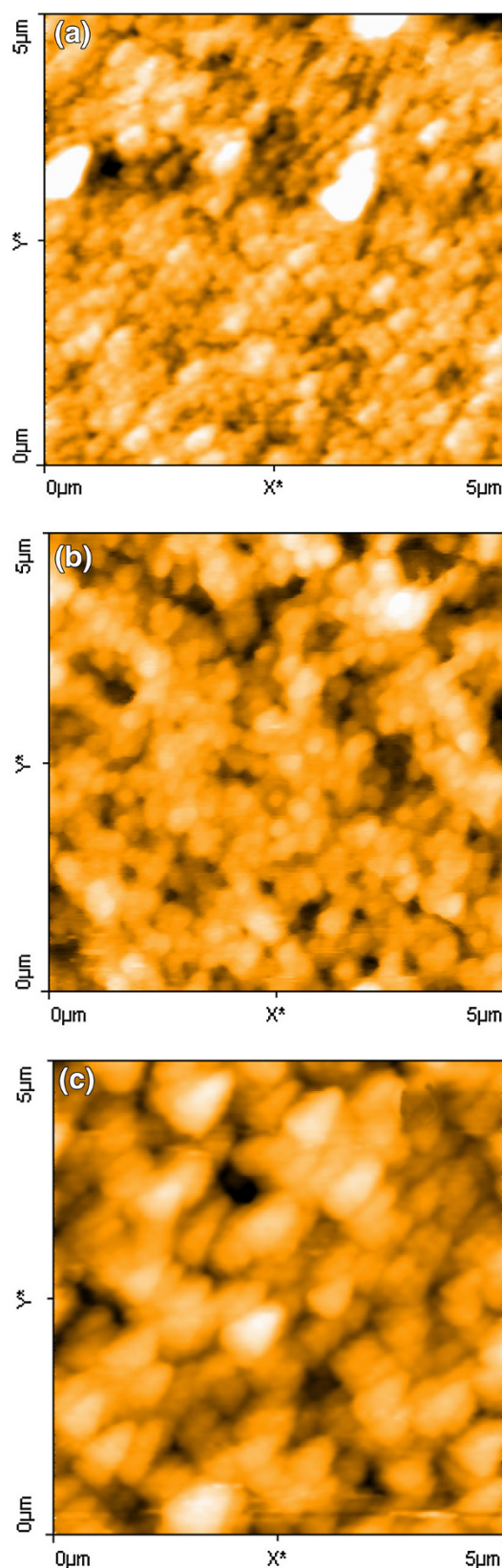
Fourier transform infrared (FTIR) spectroscopy is a promising method for observing molecular vibrations. Figure 4 shows the FTIR spectra of TiO<sub>2</sub> thin films annealed at 400, 450 and 500 °C. The thin films annealed at 400, 450 and 500 °C have peaks above 3,500 cm<sup>-1</sup> which represents the O–H stretching of hydrogen bonds on the anatase TiO<sub>2</sub> surface. A reaction with a hydroxyl anion is thought of as the initial step of the photo oxidation of water. Since the TiO<sub>2</sub> sol layer has relatively small hardness and soft surface, it seems to be easy to dissociate the water or oxygen molecules from the sol surface. The hydrate water vibration peak at 1,550 cm<sup>-1</sup> is still observed for all the three temperatures. This may be related to water bound to TiO<sub>2</sub> or may be due to the presence of reduced oxidant salt species or TiO<sub>2</sub> in a hydrate form (Xiaofeng et al. 2006). The films annealed at 400 °C exhibits a band at 3,718 cm<sup>-1</sup> which is not observed for the films annealed at 450 and 500 °C, respectively, and this is



**Fig. 4** The FTIR spectra of TiO<sub>2</sub> thin film for different temperatures

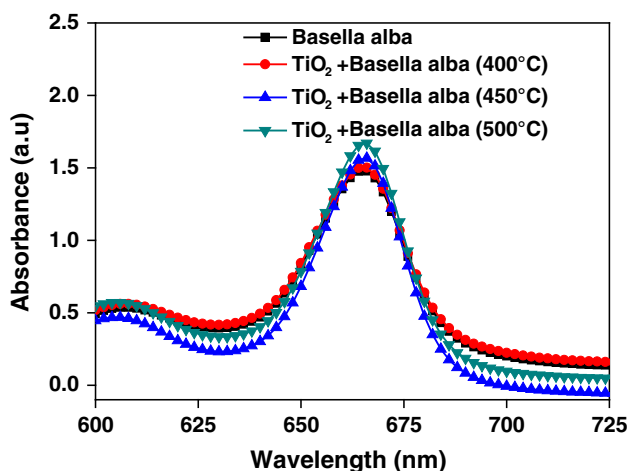
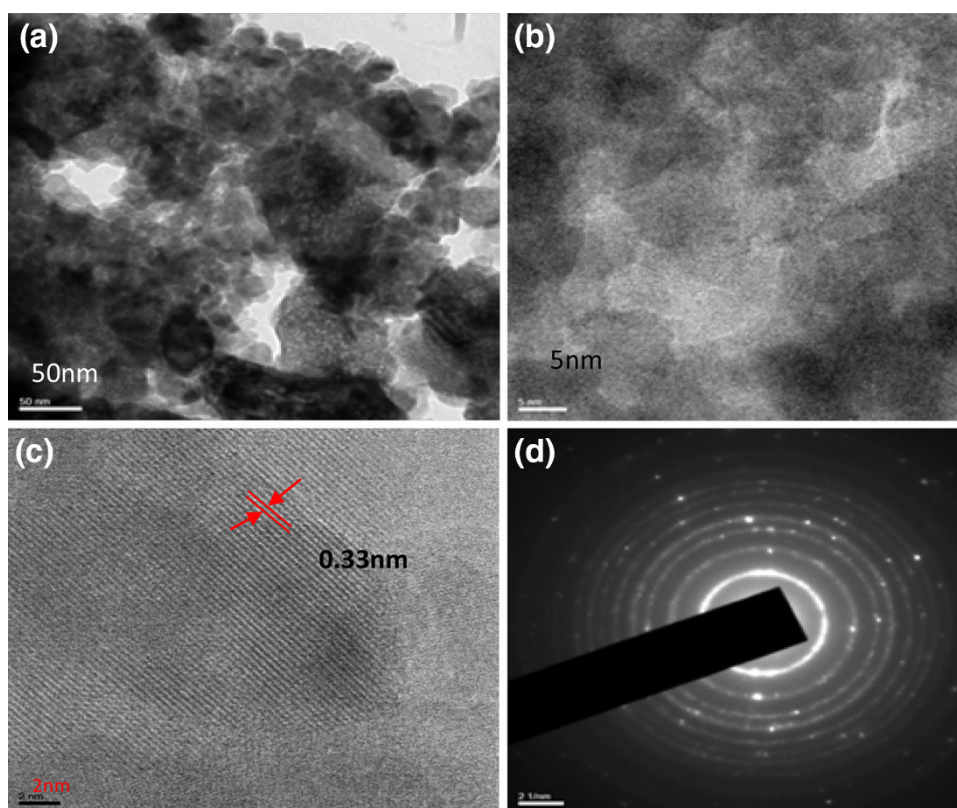
due to the removal of hydroxide group after annealing at higher temperatures. The intense band below 1,550 cm<sup>-1</sup> is due to Ti–O–Ti vibrations. The slight shift of the band to the lower wave numbers and sharpening of the Ti–O–Ti band on annealing are may be due to the increase in size of the nanoparticles. In addition, the surface hydroxyl groups in TiO<sub>2</sub> increase with the increase of annealing temperature. Also, there are no additional bands present in the spectra corresponding to the alkoxy groups. This reveals that the addition of acetic acid has not introduced any residual impurities on the TiO<sub>2</sub> surface.

The atomic force microscope images of the prepared TiO<sub>2</sub> films annealed at 400, 450 and 500 °C are shown in Fig. 5a–c. The roughness of the TiO<sub>2</sub> film is found to be 18.8, 26.6 and 31.8 nm for the films annealed at 400, 450 and 500 °C, respectively. The mesoporous structure of TiO<sub>2</sub> film annealed at 500 °C was clearly observed from the AFM image. Figure 6a shows the closely packed agglomeration of the uniform nanoparticles in the mesoporous structure, which displays the even monodispersity of mesopores. This accumulation of nanoparticles creates narrow channels that may serve as electronic injection membranes (Cao et al. 2009). It can be seen from Fig. 6b that the size of the nanoparticles is extremely uniform. Figure 6c shows lattice fringes and the interplanar distance is measured to be 0.33 nm which corresponds to the (101) lattice plane of anatase phase of nanocrystalline TiO<sub>2</sub> thin films. Selective area electron diffraction pattern is used to learn about the crystal properties of a particular region. Figure 6d shows the selective area electron diffraction pattern of the nanocrystalline TiO<sub>2</sub> thin film and the presence of rings with discrete spots suggests that the TiO<sub>2</sub> nanocrystalline film is made of small particles of uniform size (Zheng et al. 2001).



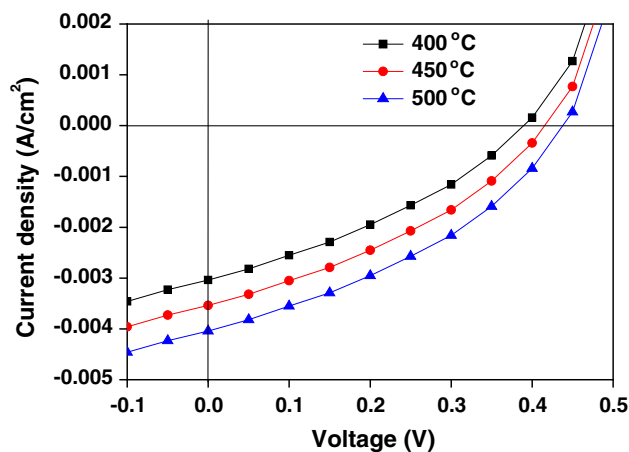
**Fig. 5** AFM images of TiO<sub>2</sub> thin films annealed at **a** 400 °C, **b** 450 °C and **c** 500 °C

**Fig. 6** HRTEM micrograph of  $\text{TiO}_2$  nanocrystalline thin films annealed at 500 °C



**Fig. 7** Absorption spectra of basella alba rubra spinach-sensitized  $\text{TiO}_2$  thin films for different temperatures

Optical absorption spectra of basella alba rubra and basella alba rubra-absorbed  $\text{TiO}_2$  nanocrystalline thin films are shown in Fig. 7. It is seen that the absorption peak of basella alba rubra extract is maximum at 665 nm. The presence of the absorbance peak at 665 nm in basella alba rubra-sensitized  $\text{TiO}_2$  nanocrystalline thin films confirms the incorporation of dye into the  $\text{TiO}_2$  film. It is seen that absorption coefficients of basella alba rubra extract are about 19 times higher



**Fig. 8** J–V characteristics of dye-sensitized  $\text{TiO}_2$ -based solar cells for different temperatures

than that of the N-719 dye in high-efficiency dye-sensitized solar cells (Senthil et al. 2011). The intensity of light absorption has been enhanced due to the interfacial Ti–O coupling which exists between the C=O, C–H, =CH<sub>2</sub>, O–H bonding of dye molecules and  $\text{TiO}_2$  molecules.

The J–V characteristics of  $\text{TiO}_2$  nanocrystalline thin films sensitized with natural dyes are shown in Fig. 8. The solar cell parameters, fill factor (FF) and efficiency ( $\eta$ ), have

been calculated using the following equations (Thambidurai et al. 2013, 2014, 2011)

$$FF = J_m \times V_m / J_{sc} \times V_{oc}$$

$$\eta = (FF \times J_{sc} \times V_{oc})$$

where  $J_{sc}$  is the short circuit photocurrent density ( $\text{mA cm}^{-2}$ ),  $V_{oc}$  is the open circuit voltage (volts),  $P_{in}$  is the intensity of the incident light ( $100 \text{ W cm}^{-2}$ ) and  $J_m$  ( $\text{mA cm}^{-2}$ ) and  $V_m$  (volts) are the maximum current density and voltage in the J–V curve, respectively, at point of maximum power output. The dye-sensitized solar cells with  $\text{TiO}_2$  prepared at  $400 \text{ }^\circ\text{C}$  show the power conversion efficiency ( $\eta$ ) of 0.38 %, a short circuit current density ( $J_{sc}$ ) =  $3.06 \text{ mA cm}^{-2}$ , open circuit voltage ( $V_{oc}$ ) = 0.39 V and fill factor (FF) = 0.32. The dye-sensitized solar cell fabricated using  $\text{TiO}_2$  prepared at  $450 \text{ }^\circ\text{C}$  achieved a power conversion efficiency ( $\eta$ ) of 0.50 % with a short circuit current density ( $J_{sc}$ ) =  $3.56 \text{ mA cm}^{-2}$ , open circuit voltage ( $V_{oc}$ ) = 0.41 V and fill factor (FF) = 0.34. The dye-sensitized solar cells with  $\text{TiO}_2$  prepared at  $500 \text{ }^\circ\text{C}$  exhibited a power conversion efficiency of 0.70 % with a short circuit current density ( $J_{sc}$ ) of  $4.35 \text{ mA cm}^{-2}$ , open circuit voltage ( $V_{oc}$ ) of 0.48 V and fill factor (FF) of 0.35. It was found that basella alba rubra extract possesses better photosensitization effect than betacyanin pigments from beets (Thambidurai et al. 2011).

## Conclusion

$\text{TiO}_2$  nanocrystalline thin films have been prepared by a simple sol–gel method. X-ray diffraction analysis reveals that the  $\text{TiO}_2$  nanocrystalline thin films exhibit anatase structure. The particle size seems to decrease with the increase of concentration of the precursor. The dyes extracted from basella alba rubra absorb visible light and have been found to be suitable for the use as sensitizer in solar cells. The efficiency of the fabricated dye-sensitized solar cell using basella alba rubra extract is 0.70 %.

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